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	Organization	Suresh Gyan Vihar University Jaipur				
	Address	Jaipur, India				
	Division					
	Organization	Kuwait College of Science and Technology				
	Address	Doha Area, 7th Ring Road, 13133, Safat, Kuwait				
	Phone					
	Fax					
	Email	r.nair@kcst.edu.kw				
	URL					
	ORCID					
Author	FamilyName	Mahdi				
	Particle					
	Given Name	Jaffer				
	Suffix					
	Division					
	Organization	Kuwait College of Science and Technology				
	Address	Doha Area, 7th Ring Road, 13133, Safat, Kuwait				
	Phone					
	Fax					
	Email					
	URL					
	ORCID					
Author	FamilyName	Amjad				
	Particle					
	Given Name	Omar				
	Suffix					
	Division					
	Organization	Kuwait College of Science and Technology				
	Address	Doha Area, 7th Ring Road, 13133, Safat, Kuwait				
	Phone					
	Fax					
	Email					
	URL					
	OKCID					
Author	FamilyName	Maree				
	Particle					
	Given Name	Kareem				
	Suffix					
	Division					
	Organization	Kuwait College of Science and Technology				
	Address	Dona Area, /In King Koad, 13133, Safat, Kuwait				
	Phone					
	Fax					
	Email					
	URL					
	ORCID					

Author	FamilyName Particle	Jain
	Given Name Suffix	Ankur
	Division	Centre for Renewable Energy and Storage
	Organization	Suresh Gyan Vihar University
	Address	Jaipur, India
	Phone	
	Fax	
	Email	ankur.j.ankur@gmail.com
	URL	
	ORCID	
Author	FamilyName	Al-Dubai
	Citizen Norma	Abund
	Suffix	Anneu
	Division	School of Engineering and the Built Environment
	Organization	Edinburgh Nanier University
	Address	Edinburgh FH10 5DT UK
	Phone	,,,,
	Fax	
	Email	
	URL	
	ORCID	
Author	FamilyName	Jaradat
	Particle	
	Given Name	Suha
	Suffix	
	Division	School of Engineering and the Built Environment
	Organization	Edinburgh Napier University
	Address	Edinburgh, EH10 5D1, UK
	Phone	
	Fax	
	URL	
	ORCID	
Author	FamilyName	Shyha
	Particle	
	Given Name	Islam
	Suffix	
	Division	School of Engineering and the Built Environment
	Organization	Edinburgh Napier University
	Address	Edinburgh, EH10 5D1, UK
	Phone	
	Fax	
	ORCID	
Author	Formily Norma	Turbalei
1 141101	Particle	1140.131
	Given Name	Mohamed
	Suffix	
	Division	
	Organization	Kuwait College of Science and Technology
	Address	Doha Area, 7th Ring Road, 13133, Safat, Kuwait
	Phone	
	Fax	
	Email	
	URL	
	ORCID	

Author	FamilyName	Alothamn
	Particle	
	Given Name	Basil
	Suffix	
	D	
	Division	
	Organization	Kuwait College of Science and Technology
	Address	Doha Area, 7th Ring Road, 13133, Safat, Kuwait
	Phone	
	Fax	
	Fmail	
	LIPI	
	ORCID	
	OKCID	
Author	FamilyName	Shehata
	Portiolo	
	Given Name	Nader
	Suffix	
	Division	
	Organization	Kuwait College of Science and Technology
	Address	Doha Area. 7th Ring Road. 13133. Safat. Kuwait
	Division	School of Engineering and the Built Environment
	Organization	Euroburgh Napler University
	Address	Edinburgh, EH10 5DT, UK
	Division	Center of Smart Materials, Nanotechnology and Photonics (CSMNP), Smart CI Research Center
	Organization	Alexandria University
	Address	Alexandria, 21544, Egypt
	Division	Department of Engineering Mathematics and Physics Eaculty of Engineering
	Organization	Alavandria University
	Address	Alexandra, 21544, Egypt
	Division	USTAR Bioinnovations Center, Faculty of Science
	Organization	Utah State University
	Address	Logan, UT, 84341, USA
	Division	School of Engineering
	Organization	Ulter University
	Adduces	
	Address	Delast, D113 IAP, UK
	Phone	
	Fax	
	Email	n.shehata@kcst.edu.kw; n.shehata@ulster.ac.uk
	URL	
	ORCID	
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Abstract	In this study, both the op	tical and piezoelectric properties of polyvinylidene fluoride (PVDF) electrospun nanofibers were investigated at
	different needle-to-collect	tor distances of the electrospinning process at constant applied high voltage. For piezoelectric characterization, the
	fabricated nanofiber mats	were subjected to applied forces, including cyclic force, variable frequency-based loads, and free-falling masses
	(impulse loading), along v	vith power density analysis for different load resistance values. In addition, both optical absorbance and transmittance
	measurements were cond	ucted to evaluate the optical properties of the fabricated nanofibers. The piezoelectric analysis demonstrated the best
	piezoresponse of the fabi	ricated nanomats at a needle-to-collector distance of 15 cm and high voltage of 22 kV. However, a trade-off between
	piezoelectric response an	d optical transmissivity was observed based on the electrospinning distance parameter. The relatively higher optically
	transparent sample exhibit	ited only moderate piezoelectric response, while the less transparent sample displayed the highest piezoelectric
	activity. Based on the op	timized sample and piezoelectric analysis, the synthesized nanofiber mat was subjected to applied mechanical stress
	in the form of variable vel	locity and momentum loads. A maximum potential of approximately 16 V was harvested through velocity and
	momentum impact, espec	cially with the addition of a double-layer PVDF membrane. Furthermore, the sensing effect of airflow pressure on
	single/double-layer PVDF	F was studied. The single-layer PVDF membrane generated 79 mV under an airflow speed of 21 km/h, while the
	double-layer membrane p	roduced 114 mV potential under the same airflow. This study highlights the diverse applications of PVDF nanofiber
	mats as multifunctional s	ensors and energy harvesting applications from mechanical shocks and airflow impact.
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ORIGINAL RESEARCH ARTICLE



Parametric Study on PVDF Electrospun Nanofibers: Optical Characteristics, Piezoelectric Analysis, and Correlated Applications

⁴ Remya Nair^{1,2} · Jaffer Mahdi² · Omar Amjad² · Kareem Maree² · Ankur Jain³ · Ahmed Al-Dubai⁴ · Suha Jaradat⁴ ·
 ⁵ Islam Shyha⁴ · Mohamed Trabelsi² · Basil Alothamn² · Nader Shehata^{2,4,5,6,7,8}

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8 Abstract

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AQ1 In this study, both the optical and piezoelectric properties of polyvinylidene fluoride (PVDF) electrospun nanofibers were 10 investigated at different needle-to-collector distances of the electrospinning process at constant applied high voltage. For 11 piezoelectric characterization, the fabricated nanofiber mats were subjected to applied forces, including cyclic force, vari-12 able frequency-based loads, and free-falling masses (impulse loading), along with power density analysis for different load resistance values. In addition, both optical absorbance and transmittance measurements were conducted to evaluate the opti-AQ2 14 cal properties of the fabricated nanofibers. The piezoelectric analysis demonstrated the best piezoresponse of the fabricated 15 nanomats at a needle-to-collector distance of 15 cm and high voltage of 22 kV. However, a trade-off between piezoelectric 16 response and optical transmissivity was observed based on the electrospinning distance parameter. The relatively higher 17 optically transparent sample exhibited only moderate piezoelectric response, while the less transparent sample displayed 18 the highest piezoelectric activity. Based on the optimized sample and piezoelectric analysis, the synthesized nanofiber mat 19 was subjected to applied mechanical stress in the form of variable velocity and momentum loads. A maximum potential of 20 approximately 16 V was harvested through velocity and momentum impact, especially with the addition of a double-layer 21 PVDF membrane. Furthermore, the sensing effect of airflow pressure on single/double-layer PVDF was studied. The single-22 layer PVDF membrane generated 79 mV under an airflow speed of 21 km/h, while the double-layer membrane produced 23 114 mV potential under the same airflow. This study highlights the diverse applications of PVDF nanofiber mats as multi-24 functional sensors and energy harvesting applications from mechanical shocks and airflow impact.

Keywords Piezoelectric energy harvesting · electroactivity · nanofibers · sensors · nanotechnology · nanogenerator ·
 transduction

			_		
A1 A2		Remya Nair r.nair@kcst.edu.kw	4	School of Engineering and the Built Environment, Edinburgh Napier University, Edinburgh EH10 5DT, UK	A13 A14
A3 A4 45		Ankur Jain ankur.j.ankur@gmail.com Nader Shehata	5	Center of Smart Materials, Nanotechnology and Photonics (CSMNP), Smart CI Research Center, Alexandria University, Alexandria 21544, Egypt	A15 A16 A17
A6	1	n.shehata@kcst.edu.kw; n.shehata@ulster.ac.uk	6	Department of Engineering Mathematics and Physics, Faculty of Engineering, Alexandria University,	A18 A19
A7 A8		Jaipur, Jaipur, India	7	USTAR Bioinnovations Center, Faculty of Science, Utah	A20 A21
A9 A10	2	Kuwait College of Science and Technology, Doha Area, 7th Ring Road, 13133 Safat, Kuwait	8	State University, Logan, UT 84341, USA	A22
A11 A12	3	Centre for Renewable Energy and Storage, Suresh Gyan Vihar University, Jaipur, India		UK	A23 A24

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surface area and suppressing repulsive forces. As voltage increases, the repulsive force becomes unstable, ejecting a charged jet from the conical drop tip.^{8,9}

Electrospun nanofibers are in high demand due to their applications in filtration, smart textiles, protective clothing, cosmetics, sensors, energy harvesting, electrical, and biomedical fields.¹⁰ A variety of piezo materials are studied for electrospun nanofibers, especially for application in the fields of sensing and energy harvesting. Polymers are a common alternative for piezoelectric conversion because of their versatility, low cost, variety of manufacturing techniques, light weight, and recyclability in comparison with other piezomaterials such as crystals and ceramics.^{11,12}

The unique phenomenon known as piezoelectricity is 66 the production of electrical energy from mechanical energy 67 without the need for an external voltage input. The elemental 68 structure connected to the crystal lattice is disrupted when 69 an external mechanical force is applied to piezomaterials. 70 The positive and negative charges are kept in a balanced 71 equilibrium in the case of a typical crystalline structure. 72 When a mechanical stress is applied, the balanced equilib-73 rium is disturbed, leading to an uneven dipole configuration 74 and the subsequent creation of electrical charge. This occurs 75 by the deformation of the internal crystal structure and dis-76 ruption of charge balance, and as a result, positive and nega-77 tive charges are separated, and the neutrality of the material 78 is destroyed. This phenomenon, referred to as spontane-79 ous polarization, is caused by the displacement of electron 80



Fig. 1 Electrospinning experimental set up used for the fabrication of nanofiber membrane.

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Introduction

Nanotechnology involves engineering active systems at a

scale of 10^{-9} m, enhancing the surface-to-volume ratio and

interatomic interaction. It is highly sought after in material

science, industrial sectors, and research areas. Nanostruc-

tures such as nanoparticles, nanowires, nanofibers, and

nanolayers have several fascinating properties that make

them the foundation for any nanotechnology-based appli-

cations.¹⁻⁴ One-dimensional nanomaterials with nanoscale

diameter or thickness, especially nanofibers, have been

extensively studied due to their captivating properties such

as excellent mechanical strength, high porosity with small

pore size, easy reusability, and flexibility.^{5,6} Nanofiber

production involves natural and synthetic materials, with

polymers being the most popular due to their flexibility,

recyclability, low cost, and affordability. Techniques for

fabrication include phase separation, template synthesis, drawing, electrospinning, and self-assembly. Advance-

ments in polymer science and chemistry have led to the

generation of synthetic nanofibers.⁷ Electrospinning is

the most cost-effective and simple method for developing

efficient nanofibers. This non-interrupted process involves

electrifying liquid droplets under a kilovolt electric poten-

tial, resulting in elongated and stretched jets, forming

ultrathin, fine nanofibers as shown in Fig. 1. Electrostatic

repulsion distorts droplets into Taylor cones, increasing

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clouds relative to their individual atomic centers, that is, 81 displacement of positive ions relative to negative ions within 82 their crystal cells producing electric dipoles. The abovemen-83 tioned displacement is called electric charge density dis-84 placement (D). The performance of a piezoelectric material 85 mainly depends on the associated electric dipoles and their 86 symmetry. Non-centrosymmetric crystals are more suscepti-87 ble to the piezoelectric effect, and they exhibit piezoelectric 88 responses upon application of an external mechanical stress. 89 The reverse of this effect occurs when an applied electrical 90 voltage results in mechanical deformation, and this effect is 91 called the converse piezoelectric effect. Sensors and trans-92 ducers are based on the principle of the direct piezo effect, 93 and actuators operate under the principle of the converse 94 piezo effect. 95

Direct piezoelectric effect generated across a material can
be expressed by the following equation,

$$\begin{array}{l} 98\\99 \end{array} \quad D = d \times T + \varepsilon \times E, \tag{1}$$

Here, the electric charge density displacement is repre-100 sented by D, T represents the applied mechanical stress, ϵ 101 represents the material permittivity, the piezoelectric coeffi-102 cient (piezoelectric charge constant) is represented by d, and 103 the electric field is represented by E. A higher voltage output 104 can be generated for a material with higher d value for the 105 same amount of mechanical deformation. This value clearly 106 indicates the piezo activity of the material. The piezoelectric 107 charge constant d is defined as the polarization produced for 108 unit mechanical stress applied on the material. This piezoe-109 lectric charge constant represents a third rank tensor where it 110 couples between a second rank tensor or vector for stress and 111 a first rank tensor for generated electric potential. Piezoelec-112 tric coefficient(d_{ii}) is an anisotropic physical quantity and 113 thus in general it relates input parameters (applied mechani-114 cal stress) to output parameters (direction of polarization or 115 electric displacement) associated with piezoelectricity using 116 double subscript. Piezoelectric charge coefficient is a third 117 rank tensor that can be expressed as 3×6 matrix that cor-118 relates charge displaced unit area associated with an applied 119 stress according to the relation, 120

$$\begin{array}{cc} 121 & D_i = d_{ij}T_j \\ 122 \end{array}$$

where d_{ii} represents piezoelectric charge coefficient (C//N), 123 D_i represents dielectric displacement (C/m²), with *i* denoting 124 the direction of charge motion, and T is the applied stress, 125 with *j* denoting the direction of applied stress. Thus, d_{33} 126 relates the ratio of charge motion along the third axis (z) 127 to the stress applied along the third axis (z), assuming the 128 electrodes shorted and no other stress is present. Stress rep-129 resents the applied force acting over the surface area of a 130 material. In other words, we can say that d_{33} means induced 131 polarization along the z-axis per unit stress applied along 132

the *z*-axis. Here, subscripts 1, 2, and 3 are used for *x*, *y*, and *z* directions, and shear about these axes is represented by 4, 5, and 6. In the case of d_{33} , the directions of polarization and applied force are parallel to each other. The d_{33} value of a material clearly indicates the relationship between crystallization and piezoelectricity.^{13,14} 133 134 135 136 137 138

Materials generating a piezo effect that can be easily elec-139 trospun into nanofibers mainly include polymers. Polymers 140 that exhibit a piezo effect include natural and synthetic types. 141 Some of the natural organic materials that exhibit a piezo 142 effect include gelatin ($d_{33} = 20 \text{ pC/N}$), silk ($d_{33} = 38 \text{ pC/N}$), 143 and PLLA ($d_{33} = 27 \text{ pC/N}$). Poly(L-lactide) (PLLA) is a pie-144 zoelectric polymer with adequate piezoresponse obtained by 145 the ring opening polymerization of an L-lactide monomer. In 146 addition to these, cellulose and chitin also have appreciable 147 piezo activity with less mechanical stability.¹⁵ Some of the 148 synthetic semicrystalline polymers that exhibit piezoelectric 149 properties mainly include polyvinylidene fluoride (PVDF; 150 $d_{33} = 20-40 \text{ pC/N}$) and its copolymers trifluoethylene, nylon 151 66, and poly ureas.¹⁶ A variety of amorphous polymers also 152 exhibit piezoelectric behavior and this category includes pol-153 vvinyl chloride (PVC), polyacrylonitrile (PAN), and polyvi-154 nyl acetate (PVAc). The main attractive properties that pro-155 mote these polymers for piezoelectric applications include 156 their high impact resistance, low acoustic and mechanical 157 impedance, high dielectric breakdown, and operating field 158 strength with low density and elastic stiffness.^{17,18} 159

PVDF $[(C_2H_2F_2)_n]$ is a semicrystalline thermoplastic 160 fluoropolymer with long-chain molecules produced by the 161 polymerization of the monomer units vinylidene difluoride 162 $(CH_2=CF_2)$. It is a highly stable non-reactive polymer with 163 around 50-60% crystallinity.¹⁹ PVDF typically exists in 164 five crystalline phases, namely α , β , Υ , δ , and ε , due to its 165 polymorphic nature. Here, α , β , and γ are the more abun-166 dant polymorphic phases, out of which β is the main focus 167 of attraction due to its piezoelectric nature. The β phase is 168 the only phase responsible for the piezoresponse of PVDF 169 polymer. The CH₂ and CF₂ groups are found in different 170 spatial arrangements along the polymer chain, and this 171 results in the generation of several fascinating properties 172 of PVDF. PVDF includes around 3 wt.% of hydrogen and 173 59.4 wt.% of fluorine, and this fluorine content greatly 174 affects the chemical and electrical properties of PVDF. 175 The presence of electronegative fluorine and electroposi-176 tive hydrogen results in the generation of dipole moment 177 inside the molecular units of PVDF. An orthorhombic 178 crystallographic system with all-trans or TTTT molecu-179 lar conformation is exhibited by the β polymorph, but α 180 polymorph exhibited a monoclinic crystallographic system 181 with a TGTG' molecular conformation. A T3GT3G molec-182 ular conformation with an orthorhombic crystallographic 183 system is shown by γ polymorph. The polar and antipolar 184 analogue of the α phase is represented by δ and ε phases. 185

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The all-trans TTTT planar zigzag molecular conforma-186 tion of the β phase results in the generation of a non-zero 187 net dipole moment, and the C-F dipoles and C-C chain 188 backbone are arranged in such a way that they cancel each 189 other out. The molecular chain arrangement of the PVDF 190 polymer and the original image of an electrospun PVDF 191 nanomembrane are shown in Fig. 2.^{20,21} 192

The piezoelectric nature of the PVDF nanofiber gener-193 ates a piezoelectric response, making it suitable for energy 194 harvesting.²²⁻²⁴ The process of turning wasted or "ambient" 195 energy into useable energy for everyday activities is known 196 as energy harvesting. Although generated power is low for 197 many electrical appliances, it is useful for low-power elec-198 tronic devices.^{25–28} Energy harvesting has various applica-199 tions such as noise barriers that harness ambient acoustic 200 energy, kinetic energy generators that harness the energy of 201 mild vibrations from common appliances such as ovens and 202 washing machines, and other related uses.^{28,29} The PVDF 203 nanofiber is a fantastic material for this application because 204 of its qualities. Of all the piezoelectric polymers, it is lead-205 free, flexible, and has one of the highest piezoelectric coef-206 ficients (d33) of roughly 12 to 35 pC/N.^{29,30} 207

PVDF nanofibers are used in energy harvesting and smart 208 systems as tactile sensors. They detect and measure body 209 signals such as respiratory cycles and heart rate in medical 210 applications. PVDF's fast response time and high sensitivity 211 make it ideal for recording pulses. It is also used in teleop-212 eration, surgeries, and complex systems to detect damage 213 in vital structural parts.³¹⁻³⁷ PVDF sensors can detect dam-214 age or cracks in components by measuring strain changes. 215 In addition, their self-excitation property makes them very 216 cost effective for many applications. High sensitivity allows 217 them to accurately detect contact force and tactile "smooth-218 ness" of materials, similar to the tactile sense of fingers.³⁸ 219 PVDF sensors are used for flow velocity measurements in 220 medical research, particularly in sleep apnea. The force of 221 gas on the nanofiber depends on airflow velocity, similar to 222 pneumotachographs. However, more research is needed on 223

the performance of electrospun PVDF nanofibers and the 224 combination of optical and electronic characterizations.^{39–45} 225

Therefore, this work presents a detailed parametric study 226 of PVDF nanofiber membranes, fabricated at different nee-227 dle-to-collector distances and consequently different applied 228 electric fields. The generated nanofibers are tested through 229 piezoelectricity measurements to detect the optimized elec-230 trospinning process parameters for the best piezoelectric 231 performance. In addition, the impact of varied processing 232 parameters is investigated with respect to the surface mor-233 phology of the formed PVDF structures, whether in thin 234 film or nanofiber mat form, while also assessing their optical 235 properties and piezoelectric response. To identify the most 236 effective nanomembrane from the piezoelectric perspective, 237 both voltage and power density measurements are performed 238 under different mechanical excitations. In addition, the mor-239 phological characterization of the generated samples is ana-240 lyzed using scanning electron microscopy (SEM) images, 241 while the chemical composition and beta sheet formation 242 are analyzed via Fourier transform infrared (FTIR) spec-243 troscopy. As part of application-based studies, this study 244 applies perturbations in the form of variable momentum 245 and airflow rates to the sample with the best piezoelectric 246 response. These findings will be useful in the application of 247 energy harvesting or airflow sensing to detect gas leakages 248 in complex systems or for generating electricity using wind. 249

Experimental Work

Materials

Polyvinylidene difluoride (PVDF; Sigma Aldrich) and 252 Kynar[®] (Arkema, King of Prussia, PA, USA) were the main nanofiber membranes used in the present study. The solvent used for the preparation of PVDF solution was dime-255 thyl formamide (DMF) (98%, Sigma Aldrich, Taufkirchen, 256 Germany). To prepare the solution, 10 g PVDF powder 257



Fig. 2 (a) PVDF chain configuration, (b) A sample of PVDF electrospun nanofiber membrane on an aluminum foil sheet.

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was dissolved in 10 ml DMF and left on a magnetic stirrer
overnight to achieve a homogeneous solution of PVDF 10
wt.%. The prepared air bubble-free solution was loaded into
a syringe with a needle for electrospinning.

262 Electrospinning Process

A schematic of the electrospinning setup is shown in Fig. 1. 263 The plastic syringe with a stainless-steel 18-gauge needle 264 was loaded with the required solution and then was fixed 265 onto an injection pump (NE1000-Single Syringe Pump, New 266 Era, Farmingdale, NY, USA) to control the feed rate of the 267 solution during the electrospinning process. The feed rate 268 was set for a flow rate of 2 ml/h, and the distance between the 269 needle and collector was varied within a range from 12 cm 270 to 15 cm to test the effect of needle-to-collector distance on 271 the generated nanofiber membrane. A 30 kV/1.65 mA regu-272 lated high-voltage power supply (Divotech Technologies) 273 was used for the application of a constant 22 kV voltage on 274 the metallic needle, and the negative terminal of the power 275 supply was connected to the collector plate, where it was 276 finally grounded. A 15×17 cm static copper plate covered 277 with aluminum foil was used as a collector electrode for 278 concentrating the deposition of nanofibers.⁴⁶ The generated 279 nanofiber membranes were peeled off carefully from the 280 aluminum foil-covered collector electrode. A total of 10 ml 281 polymer solution was deposited onto the collector electrode 282 at a feed rate of 2 ml/h, and thus a total of 5 h was required 283 for the electrospinning process to fabricate each sample. A 284 set of four PVDF nanomembranes were fabricated by adjust-285 ing the needle-to-collector distance from 12 cm to 15 cm 286 in steps of 1 cm. All other process parameters were kept 287 constant throughout the experiment. The high-voltage power 288 supply was adjusted to maintain a voltage of 22 kV through-289 out the experiment, and the flow rate was fixed for 2 ml/h. 290 The characterization of these nanofiber membranes was 291 done to check the optimum parameters that generated the 292 best nanofiber sample. The nanofiber membranes were cut 293 into an appropriate size of 2×2 cm, and these membranes 294 were sandwiched between two aluminum foil sheets from 295 which shielded copper wires were connected. The membrane 296 with aluminum foils sheets was made into a single unit using 297 paper tape. Finally, the sample was fixed horizontally with-298 out any air gap onto the base of the custom-made setup for 299 piezo characterization in order to apply a vertical force onto 300 the sample. 301

302 Impact of Momentum on PVDF Nanofiber303 Applications

The fabricated nanofiber membrane displaying the best piezoelectric performance was cut into a piece of 2×2 cm, and the sample was sandwiched between two aluminum foil electrodes connected to insulated copper wires and then integrated inside two setups to detect both shutter velocity and
airflow speed. The best fabricated nanofiber was electrospun
at a needle-to-collector distance of 15 cm at an applied volt-
age of 22 kV with a flow rate of 2 ml/h.307
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As shown in Fig. 3, the experimental setup for inspecting 312 the impact of momentum included a frictionless air track 313 with a freely moving cart controlled by a magnet at the start-314 ing position, where additional mass could also be added. 315 The sample was positioned vertically in the path of the cart 316 with a head radius 1 cm. The cart hit the nanofiber mem-317 brane in a horizontal direction at different applied velocities, 318 and the impacts resulted in the generation of piezoelectric 319 voltage. Piezoelectric voltage that developed in response to 320 the applied velocities were recorded via an oscilloscope by 321 connecting the insulated shielded wires to both aluminum 322 foils. The experiment was performed three times. First, the 323 impact of different applied velocities ranging from 0.1 m/s to 324 0.5 m/s was investigated on the first sample, after which the 325 same procedure was repeated by changing the first sample 326 with a second sample. Piezoelectric voltage generated for 327 the second sample was also recorded. Finally, the two sam-328 ple pieces were attached together by removing the insulated 329 wires on one side of the samples and then attaching them 330 together using the conducting aluminum foil sheets such that 331 the double-layer sample acted as a single piece. The piezo-332 electric voltage generation for this third sample, which was 333 the combination (series-like combination) of the two sample 334 pieces, was studied. The arrangement of the single-layer and 335 double-layer nanomats between the conducting electrodes is 336 represented in detail in Fig. 3. Two single layers are attached 337 in such a way that the bottom portion of the second layer is 338 directly attached to the top portion of the first layer with-339 out any air gap, and thus a series connection is maintained 340 between them. 341

Airflow Sensing Studies of the Fabricated Nanofiber 342 Applications 343

The setup used to create the airflow, as shown in Fig. 4, 344 consists of an electric air duster with an attached nozzle (1 345 cm radius), which was pointed at the center of the PVDF 346 sandwiched unit. In order to change the speed of the air-347 flow from the air duster, the distance was varied for each 348 speed, which was measured using a velocity meter placed 349 on a table and a ruler to measure the distance between the 350 air duster's nozzle tip and the fan blade of the velocity 351 meter. For each airflow speed value, the air duster was 352 held at the corresponding height and pointed at the center 353 of the sample. The sample was exposed to the airflow for 354

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Fig. 3 Experimental setup used for the momentum and velocity response analysis.



Fig. 4 Experimental setup used for PVDF airflow sensing-based analysis.

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1 s, and the average was calculated by taking six different
peak-to-peak voltage readings from the given signal on
the oscilloscope.

Characterization of the Fabricated Nanofiber Membrane

360 **Optical Characterization**

The optical characterization of the set of four samples 361 fabricated at different needle-to-collector distances was 362 performed using a UV-Vis spectrophotometer (Shimadzu 363 UV-2600). The optical characterization includes the gener-364 ated nanofiber absorbance and transmittance analysis. Here, 365 the sample to be analyzed was illuminated by a light source 366 in the wavelength range from 190 nm to 900 nm. The experi-367 mental setup mainly includes a light source, a monochro-368 mator or a prism to separate wavelengths, a sample holder, 369 and a detector. Transmittance and absorbance measurements 370 were based on the comparison of light intensity before and 371 after passing through the sample to be tested. Transmittance 372 can be defined as the ratio I/I_0 where I represents the inten-373 sity of the light after passing through the sample and I_0 rep-374 resents the full intensity of light before passing through the 375 sample. The absorbance is the amount of light absorbed out 376 of the incident light by the sample. The optical activity of the 377 generated sample indicates the potential of these membranes 378 in the field of light-based or transparency-based applications 379 such as touch panels and transparent wearable sensors. 380

381 Chemical Characterization

FTIR spectroscopy was used to analyze the response of these 382 nanofibers towards the piezo activity based on the beta sheet 383 content. The molecular fingerprint of the sample can be 384 easily analyzed using FTIR spectra based on the functional 385 groups present and the interactions between them. The trans-386 mittance spectra of the PVDF nanofibers were generated 387 using a PerkinElmer FTIR spectrometer at a scanning resolu-388 tion of 5 cm⁻¹ around a range of 4100–400 cm⁻¹. Each sam-389 ple was scanned about 120 times while generating the trans-390 mittance spectra. Determining the chemical structure of a 391 sample is mainly based on the molecular vibrations. Stretch-392 ing and bending are the two types of molecular vibrations 393 involved in IR spectroscopy. The stretching mode changes 394 the bond length, resulting in either symmetric or asymmetric 395 mode, whereas the bending mode always changes the bond 396 angle. Whenever the frequency of a specific bond vibration 397 becomes exactly equal to the frequency of the applied IR 398

radiation, the radiation will be completely absorbed by the 399 sample content, generating a plot of transmittance. 400

Morphological Characterization

A nanofiber sample measuring 1 × 1 cm was analyzed using a scanning electron microscope (JEOL JSM-7610FPlus FESEM, Jaipur, India) to study the surface and morphology of the generated nanofiber mat. The nanofiber size distribution on the fabricated membrane was studied and measured using ImageJ software.

Piezoelectric Characterization

Piezoelectric characterization associated with the fabri-
cated nanofibers was performed using an applied pertur-
bation in the form of falling masses and applied forces,
and the corresponding generated voltage was measured
directly using an oscilloscope. The generated voltages
were measured across different load resistance.409
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Voltage Characterization

The piezoelectric characterization of nanofibers includes 416 different applied force and frequency measurements, as 417 presented in Fig. 5. Different compressive loads/forces 418 from 0.5 N to 3 N in steps of 0.5 N were applied onto the 419 nanofiber sample at a single frequency of 1 Hz, and the 420 corresponding generated voltage was measured. Frequency 421 measurements were performed by varying the frequency 422 between 0.25 Hz up to 10 Hz at a specific applied force 423 of 2 N and measuring the corresponding generated piezo-424 electric voltages. 425

In addition, piezoelectric characterization based on impulse loading at a height of 2 cm was also performed. Impulse loading shows the effect of masses falling onto the sample, where the mass was varied from 50 g to 250 g and the generated output voltage was measured. 426 427 428 429 429 429 429 429 429 429

Power Density Characterization

As shown in Fig. 6, the piezoelectric performance of each 432 sample based on different applied load resistance across 433 the positive and negative electrode plates was studied. Dif-434 ferent load resistance was applied across the electrodes as 435 part of the characterization. Here, the oscilloscope probe 436 was connected across the positive and negative electrode 437 plates through the load resistance. Finally, the power out-438 put obtained for each sample was calculated over the area 439 of impact to obtain power density and plotted against the 440

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Cyclic controlled impulse

Fig. 6 Experimental setup used for the power density measurements of the PVDF electrospun mat.

corresponding load resistance. The area of contact in the
present study is circular with an effective radius of around
1.4 cm. The power density plots give us an indication of the
efficiency of the fabricated nanomembrane towards some
resistive loads in terms of piezo activity.⁴⁷

Results and Discussion

Optical Characterization

Optical characterization of the set of samples was performed448using UV-Vis spectrophotometry, recording absorbance and449transmittance data. The absorbance spectra of the corre-450sponding samples are shown in Fig. 7. The nanofiber mem-451brane showed clear optical absorption in the region from452

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Fig. 7 Absorbance spectra of pure PVDF nanofiber samples.

200 nm to 400 nm, which clearly shows the absorption peak 453 of the PVDF polymer. Absorbance spectra of the samples 454 clearly indicates the optimum performance of the nanofiber 455 456 membrane fabricated at a needle-to-collector distance of 15 cm, where maximum absorption intensity is obtained. 457 We can conclude from the absorbance curve that the PVDF 458 459 content is greater in the nanofiber membrane electrospun at distances of 15 and 14 cm, with good nanofiber content, 460 whereas the remaining two samples displayed lower absorp-461 tion intensity. In the visible range, the mat electrospun at 12 462 cm exhibited scattering effects near the baseline, thereby 463 generating good absorption in the visible region compared 464 to other electrospun mats.⁴⁸ 465

Transmittance analysis of the samples is shown in Fig. 8. 466 Maximum transmittance is found for the nanofiber samples 467 fabricated at a needle-to-collector distance of 12 cm, which 468 exhibited a film-like behavior. This clearly confirms the 469 transparency of this sample. Nanofiber membranes electro-470 spun at a distance of 15 cm showed the lowest transparency, 471 with good absorbance. Nanofiber membranes developed at 472 distances of 14 and 13 cm also showed good absorbance and 473 474 low transparency. The most transparent samples exhibited film-like behavior, with less piezo activity, as will be veri-475 fied later. Optical characterization of these samples revealed 476 477 the absorbance and transmittance rate for each sample. The nanofiber membrane fabricated at a distance of 15 cm con-478 sists of a rough surface with numerous pores and solid fib-479 ers. This structure creates some opaqueness towards light, 480 resulting in weakly transparent behavior. Here, the pores are 481 interconnected, resulting in the development of dozens of 482 483 fiber-air interfaces, and in this case, interfacial adherence and packing density has to be considered, which ultimately 484 promotes the hazy appearance towards light. When consider-485 ing the internal structure, we should take into account the air 486

present inside the nanofiber which promotes the scattering 487 of light to a great extent, further reducing the transmittance. 488 All these effects are identified in the presence of fiber-like 489 structures. The film electrospun at a distance of 12 cm con-490 sists of fewer pores and fiber-like structures, resulting in 491 appreciable transparency that allows light to pass through. 492 Thus, transparency is found to decrease with an increase in 493 the feeder-to-collector distance from 12 cm to 15 cm, where 494 the fiber-like structure formation originates above 12 cm, 495 and complete development of fibers occurs at 15 cm. 48,49 496

In addition, the film cast at a distance of 12 cm con-497 sists of fewer pores and the smallest number of fiber-like 498 structures, allowing the effective transmittance of light. 499 Here, the majority of chains are formed as film without 500 successful cross-linking and interconnected networks, pro-501 moting higher transmittance through it. The transparency 502 of a polymer-based electrospun mat mainly depends on 503 its crystallinity. Materials with higher crystallinity exhibit 504 less transmittance. This is clearly evident from the FTIR 505 spectra calculation, which shows that the 12 cm-based mat 506 exhibits the lowest beta sheet content, indicating lower 507 crystallinity and thereby showing excellent transparency.⁴⁸ 508

Chemical Characterization

The crystal phases of the pure PVDF nanofiber membrane 510 are clearly projected in the IR transmittance spectra shown 511 in Fig. 9. The most prominent bands related to the crystal 512 phase of PVDF are observed in the wave number range 513 from 700 cm⁻¹ to 1500 cm⁻¹.⁵⁰ Among these, the transmit-514 tance peaks observed at 762 cm⁻¹, 796 cm⁻¹, 875 cm⁻¹, 515 976 cm⁻¹, 1210 cm⁻¹, and 1383 cm⁻¹ correspond to the α 516 phase, and peaks at 840 cm⁻¹, 880 cm⁻¹, 1175 cm⁻¹, and 517 1275 cm⁻¹ are clearly associated with the β phase. The 518 relative fraction of beta phase content can be calculated 519 from the intensity of absorbance bands at 762 cm^{-1} and 520 840 cm^{-1} using the Beer–Lambert law as shown in Eq. 2 521 given below:⁵¹ 522

$$f(\beta) = \frac{A_{\beta}}{1.26A_{\alpha} + A_{\beta}} \tag{2}$$

where A_{β} represents the absorbance value at 840 cm⁻¹ and A_{α} represents the absorbance value at 762 cm⁻¹.

The absorbance peak corresponding to the α crystalline 527 phase at 762 cm^{-1} is due to the skeletal bending vibration 528 mode C(H)-C(F)-C(H) and CF_2 bending mode. Similarly, 529 the absorbance peak at 840 cm^{-1} due to the polar beta 530 phase represents a rocking vibrational band due to CH₂. 531 Based on Eq. 2, the beta phase content associated with 532 each nanomat fabricated at different needle-to-collector 533 distances is calculated and given in Table I.⁵² 534

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Fig. 8 (a–d) Transmittance of electrospun samples at different feeder-to-collector distances. (e) The letters N, A, N, O correspond to the samples fabricated at feeder-to-collector distances of 12, 13, 14, and 15 cm, respectively, from left to right.

The beta phase calculation revealed that the electroactive polar phase is maximum for the nanomat fabricated at a needle-to-collector distance of 15 cm and is found to be decreasing while reducing the distance from 15 cm down to 12 cm. The beta phase content is due to the CH_2-CF_2 dipole generation inside the PVDF nanomaterial, resulting in an excellent piezoresponse.

542 Morphological Characterization

Figure 10 shows the SEM images of the electrospun mats
generated at different feeder-to-collector distances of 12 cm,
13 cm, 14 cm, and 15 cm. The electrospun mat fabricated
at a distance of 12 cm exhibited a thin film-like behavior,
as shown in Fig. 10a. Upon increasing the distance from
12 cm to 13 cm, the electrospun mat shows a tendency for

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fiber generation from the cast film with an uneven surface, 549 as shown in Fig. 10b. Further increasing the distance from 550 13 cm to 14 cm leads to the formation of more fiber-like 551 structures, as shown in Fig. 10c. In general, the tendency to 552 form fibers is improved by increasing the feeder-to-collector 553 distance. Finally, after reaching a distance of 15 cm, ultrafine 554 nanofibers are clearly observed in the SEM image, repre-555 sented by Fig. 10d. Thus, a feeder-to-collector distance of 15 556 cm resulted in the production of the best nanofibers based on 557 the morphological analysis. The increased distance between 558 the needle and collector provides an extended solvent evap-559 oration time during solution travel and deposition. Along 560 the pathway from the needle tip to the collector electrode, 561 successful fiber generation is possible only if an optimum 562 journey time is obtained. Solvent evaporation time should 563 be optimum for obtaining nanofibers with good morphology. 564

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Fig. 9 The FTIR spectra of the generated nanomats.

 Table I
 Calculated beta phase content percentage of the electrospun

 PVDF nanomats
 PVDF

Sample	$A(\alpha)$	$A(\beta)$	$f(\beta)$
15 cm	0.039976	0.187	0.7877
14 cm	0.06004	0.2237	0.7477
13 cm	0.065	0.196	0.705
12 cm	0.0785	0.2277	0.697

The numbers in the sample column refer to the distance between the feeder and collector in the electrospinning setup

However, increasing the distance above a certain limit 565 reduces the electric force of attraction, negatively affecting 566 the fiber production. This clearly shows that a feeder-to-col-567 lector distance of 15 cm is optimum, providing enough time 568 for the deposition of charged fluid onto the collector in the 569 form of fine ultrathin nanofibers after the successful removal 570 of the volatile solvent. In other cases, decreasing the distance 571 between the needle and collector affects the fiber generation 572 due to the lack of successful evaporation of the solvent from 573 the solution for fine solid nanofiber deposition. The SEM 574 575 image of the nanomembrane produced at a feeder-to-collector distance of 15 cm was analyzed using ImageJ software, 576 and the diameter of the randomly distributed nanofibers was 577 measured. The average diameter of the generated nanofibers 578 was measured to be around 112.8 nm, and the corresponding 579 fiber diameter distribution is shown in Fig. 11.53 580

Piezoelectric Characterization

Voltage Characterization

A variable force or compression load from 0.5 N to 3 N 583 was applied to the nanofiber sample using a spring-based 584 motor arrangement, which is a custom-made setup that 585 allows different vertical forces or frequencies to be applied 586 onto the surface of the sample. The response of a set of 587 four nanofibers fabricated via electrospinning at needle-588 to-collector distances of 12 cm, 13 cm, 14 cm, and 15 cm 589 was studied. Optimum piezoresponse to the variable forces 590 was shown by nanofibers at a needle-to-collector distance 591 of 15 cm. Nanofibers fabricated at a needle-to-collector 592 distance of 13 cm also showed good piezoresponse, but 593 it was less than that of the sample prepared at the 15 cm 594 distance. The force-voltage relationship developed on 595 the nanofibers is shown in Fig. 12a. The piezoresponse 596 towards the applied force or compression load showed a 597 positive result, where the piezo voltage increased with 598 increased loads. The piezoresponse of the nanofiber fab-599 ricated at a needle-to-collector distance of 15 cm showed 600 a weaker response to lower loads, strengthening for higher 601 loads, compared with other samples. The variation in the 602 AC sinusoidal voltage obtained from each nanomembrane 603 fabricated at different feeder-to-collector distances under a 604 specific applied force of 3 N is shown in Fig. 12b. In gen-605 eral, for all the nanofiber membranes, an increase in volt-606 age is observed with increasing loads. The piezoresponse 607

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Fig. 10 SEM images of the pure PVDF nanomats fabricated at a feeder-to-collector distance of (a) 12 cm; (b) 13 cm; (c) 14 cm; (d) 15 cm.



Fig. 11 Electrospun PVDF nanofiber diameter distribution curve.

of the samples to the load is indicated by a sensitivity parameter, which represents the slope of the force-voltage graph. Linear fitting of the experimental data clearly showed a piezo sensitivity of 1.2 V/N for the nanofiber sample fabricated at a needle-to-collector distance of 15 cm. Mats produced at needle-to-collector distances of 12 cm, 13 cm, and 14 cm showed a sensitivity of 0.27 V/N,

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1.15 V/N, and 0.9 V/N, respectively. Samples developed at615needle-to-collector distances of 15 cm and 13 cm showed616the most effective response.617

Out of the four electrospun mats, the electrospun nano-618 mat generated at a feeder-to-collector distance of 15 cm 619 exhibited the best piezoresponse. Here, the generation of 620 nanofibers resulted in the formation of a higher density of 621 dipoles upon excitation with applied forces. The distance 622 affects the applied electric field on the PVDF polymeric jet 623 between feeder and collector. This electric field controls the 624 formed piezoelectric properties according to their impact on 625 the alignment of electric dipoles inside the nanofibers. This 626 resulted in the generation of a higher piezoresponse voltage 627 and consequently higher beta sheet content. In the case of 628 the mat electrospun at 14 cm feeder-to-collector distance, the 629 fabrication of nanofiber-like structures are not completely 630 successful, but some fiber generation is confirmed from the 631 SEM image. Thus, lower dipole densities are obtained upon 632 excitation forces compared with mats produced at the 15 cm 633 distance. Their beta sheet contents were also found to be 634 lower than those of the nanomats produced at the 15 cm, 635 resulting in weaker responses. The mat electrospun at 13 cm 636 feeder-to-collector distance resulted in the generation of 637 larger, non-homogeneous beads and fewer fiber-like struc-638 tures compared to the mat produced at 14 cm. The formation 639 of beads results in the generation of a greater piezoresponse 640

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Fig. 12 (a) Applied force versus piezo voltage for pure PVDF nanofiber membrane. (b) An example of AC sinusoidal voltage developed at 3 N force for different feeder-to-collector distances.



Fig. 13 Applied frequency versus piezoelectric voltage for pure PVDF nanofiber membrane.

than the mat produced at 14 cm, but with a lower beta sheet
content due to the presence of a large number of non-homogeneous beads. Finally, the electrospun mat fabricated at a
12 cm distance exhibited thin cast film-like behavior and
consequently weak piezoresponse. Thus, we can conclude
that more beads were produced as a result of reducing the
TCD (tip-to-collector distance) from 15 cm to 13 cm, and a

simultaneous departure from a fiber-like structure. Higher 648 piezo voltage for the 13 cm TCD was produced owing to the 649 presence of beads in a significant section of the electrospun 650 mat. Additionally, the beta phase content is significantly 651 reduced by the non-homogeneous distribution of beads, 652 which lowers the crystallinity content. The transmittance 653 of the electrospun mat is drastically improved by a decrease 654 in crystallinity and beta sheet content, which explains why 655 the transparent electrospun mat in this work was created at 656 a lower TCD.48,54,55 657

Effects of needle-to-collector distance on the piezoresponse of nanofiber membranes are analyzed based on cyclic forces. Different cyclic forces of specific frequencies between 0.25 Hz and 10 Hz at a constant force of 2 N are applied on these samples. Frequency-based piezo analysis curves are represented in Fig. 13. The curves clearly indicate a linear response to lower frequencies from 0.5 Hz to 2 Hz and saturation for higher frequencies of 6 Hz and 8 Hz.

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Optimum piezo performance based on frequency was also shown by the nanofiber membrane fabricated at a needle-tocollector distance of 15 cm. A needle-to-collector distance of 13 cm showed appreciable behavior, but it was less than that of the sample prepared using the distance of 15 cm. In general, the piezoresponse was found to be the best and most consistent for nanofibers fabricated at a needle-to-collector distance of 15 cm.

The active piezoresponse shown by the nanofiber mem-674 brane electrospun at 15 cm compared to that electrospun 675 at 12 cm can be explained in terms of the higher surface-676 to-volume ratio and porosity of the nanofiber membrane. 677 Piezoelectric materials are commonly non-centrosymmetric 678 materials, and as a result, the application of a vertical force 679 or stress perpendicular to the sample shifts the center of 680 mass of positive and negative ions, creating a net polariza-681 tion inside the material. Thus, a voltage difference or poten-682 tial between two surfaces of the PVDF sample is generated 683 via dipole generation throughout the materials by changing 684 its dimensions. The dipoles inside the material usually can-685 cel each other out; however, the dipoles at the surface do 686 not, which produces polarity. When these samples are in 687 the form of one-dimensional nanofibers, their surface-to-688 volume ratio and porosity increase extensively compared 689 to the electrospun film counterparts. As a result, a higher 690 dipole surface-to-volume ratio is seen in the case of the 691 nanofiber membrane, resulting in the efficient extraction of 692 voltage compared to the electrospun cast film counterpart.⁵⁶ 693 The main dipoles inside the PVDF material are CH₂–CF₂ 694 , accounting for the development of an electroactive beta 695 phase and higher piezoresponse. In addition, the higher 696 surface-to-volume ratio favors the trapping of electrostatic 697 charges. Consequently, under deformation, trapped charges 698 generate voltage and current. For the case of 13 cm and 699 14 cm membranes, the fiber formation was only initiated 700



Fig. 14 Applied mass versus piezoelectric voltage for pure PVDF nanofiber membrane

and not completely developed, generating less voltage than
the 15 cm sample. Thus, the excellent porosity and higher
surface-to-volume ratio of nanofiber membranes result
in the highest piezoresponse, with significant electroactive beta phase content in the case of the 15 cm nanofiber
membrane.⁵⁷

Impulse load testing shows the effect of falling masses, 707 and the associated impact results in the generation of volt-708 age. The associated voltage curves are shown in Fig. 14. 709 which clearly show the increase in voltage generation with 710 increased falling masses. Impulse load piezo analysis was 711 performed for the set of samples, showing an appreciable 712 response for the 13 cm nanofiber samples, and maximum 713 voltage generation was found for the 15 cm nanofibers. Thus, 714 a positive correlation exists between applied mass and gener-715 ated voltage for all the samples.⁵² A maximum potential of 716 3.2 V is obtained by the application of 3 N force, and a gen-717 eration of around 4.5 V is observed through impulse load-718 ing during the application of a mass of 300 g from a height 719 of 2 cm. Cyclic force or frequency-based mechanical stress 720 721 produced a maximum force of 3.2 V at a frequency of 10 Hz.

722 Power Density Characterization

The generated piezo voltage is dependent on the load resist-723 ance connected externally across the electrodes, which is 724 used for the estimation of power. Power density calcula-725 tions for the generated samples exhibited random behavior. 726 This was exhibited by random power increases for certain 727 728 resistance loads, followed by decreasing power density output with increasing load resistance, and then again, an 729 increase in the power density output was observed for further 730 increases in the applied resistance loads. But this fluctuating 731

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Fig. 15 The plot of power density versus load resistance for the PVDF nanofiber membrane.

behavior was not present in the case of the nanofiber mem-732 brane fabricated at a needle-to-collector distance of 15 cm, 733 where a prominent Lorentzian curve was obtained for the 734 power density plot. Here, maximum power density was 735 obtained for a resistance load of $1.5 \text{ M}\Omega$, followed by a grad-736 ual decrease with an increase in the resistance load from 2.2 737 M Ω , 2.7 M Ω , 3.3 M Ω , and 3.7 M Ω , and the corresponding 738 plot is shown in Fig. 15. A maximum power density of 9.7 739 mW/m² was obtained for a potential of 3 V in the case of 740 the PVDF nanofiber mat fabricated at a distance of 15 cm.⁵⁸ 741

Impact of Momentum on the Nanofiber Membrane 742

One of the primary sources of mechanical energy present in 743 the majority of engineering application systems is mechani-744 cal shock. By altering the object's momentum and the speed 745 at which it strikes the sample, the reaction of the electrospun 746 nanomat to mechanical shocks is examined. The findings 747 reveal a favorable reaction to mechanical shocks, demon-748 strating the potential of these nanomats for energy harvest-749 ing applications involving mechanical shocks. Therefore, by 750 altering the momentum's effect on the nanomats, this paper's 751 concept is to use these sensitive nanofiber mats as a piezo 752 or vibration energy harvester to extract electrical energy 753 from mechanical shocks. The effect of applied mechanical 754 shocks on the nanomats is varied by changing the veloc-755 ity with which the frictionless cart is horizontally hitting 756 the sample. The momentum of the cart is represented as 757 the product of its mass and velocity. First, the impact of 758 different applied velocities from 0.1 m/s to 0.5 m/s on the 759 first sample piece was studied, after which the same pro-760 cedure was repeated by changing the first sample with the 761 second sample. Even though the sample pieces were taken 762 from the same nanofiber membrane sheet, the piezoresponse 763

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Fig. 16 Piezo voltage analysis based on the impact of momentum in pure PVDF nanofiber membrane.

shown for the same velocity values were different. The first 764 piece, labeled sample 1, showed a good piezoresponse in the 765 range 1.8-12.5 V compared to the second sample. The sec-766 ond sample generated less voltage in the range of 0.5–3.8 V. 767 768 The piezoelectric voltage output generated in each case is shown in Fig. 16. The combined sample, labeled as sample 769 3, showed the highest voltage output, which was almost the 770 771 sum of the voltage generated for samples 1 and 2. For both single- and double-layer formations, the mechanism of simi-772 larly aligned diploes within the two layers produced a similar 773 impact to series connections of the two layers, which led 774 to incrementally higher generated voltage compared to the 775 single-layer case, as clearly shown in Fig. 3. The series com-776 bination of the two pieces in effect showed the double-layer 777 effect, which clearly enhanced the performance of the energy 778 harvester, generating a voltage in the range of 2–16 V. This 779 clearly indicates that the addition of different layers of PVDF 780 nanofiber membrane can enhance the piezoresponse to a 781 great extent, which is promising for several applications. The 782 addition of two layers of PVDF nanofiber membrane greatly 783 enhanced the durability of the sample. Thus, the double-784 layer sample not only improved the piezoresponse, but also 785 786 greatly enhanced the mechanical stability.

Thus, the impact of momentum on these samples is studied by varying the applied velocity, and the efficiency of the
fabricated nanomats in the field of energy harvesting from
mechanical shocks is clearly proved in the present study.

Airflow Sensing Studies of the PVDF NanofiberMembrane

For each airflow speed value, the air duster was held at the
corresponding height and pointed at the center of the sample. The sample was exposed to the airflow for 1 s and the



Fig. 17 Voltage versus airflow velocity for pure PVDF nanofiber membrane.

average was calculated by taking six different peak-to-peak 796 voltage readings from the given signal on the oscilloscope. 797 Due to external factors, constant noise was shown on the 798 oscilloscope (ranging from 20 mV to 40 mV). Before 799 each measurement, the noise was first recorded and then 800 subtracted from the reading after the measurement. For 801 the double-layer PVDF sample, the original single-layer 802 sample was paired with another sample, which was pre-803 pared in the same manner and tested for its piezoresponse. 804 Each sample had a piece of aluminum foil on one side 805 (the outer side) and one shared piece of aluminum foil in 806 between. The samples were then framed with paper tape 807 and had one wire on either side, making it a series circuit. 808 Due to the experimental setup, as the distance between the 809 electric duster's nozzle and the PVDF sample increased, 810 the air was more dispersed and less directed towards the 811 sample, resulting in a lack of smoothness. This increase 812 is seen in the graphs in Fig. 17, which show an obvious 813 relationship in the graphs regardless. When the two sam-814 ples were put together to create the double-layer sample, 815 there was an increase in the output voltage of the PVDF 816 sample; however, it was not exactly equal to the sum of 817 the voltages of the two samples. This is because the layer 818 of PVDF that experiences direct impact from the electric 819 duster's airflow absorbs some of the force, meaning that 820 less of it impacts the second layer. 821

The detailed results regarding the piezo voltage generated from the samples are listed in Table II. The results in Table II show that the double-layer PVDF sample did exhibit an increase in the voltage response compared to the single-layer sample, but with significant inefficiency, especially at higher airflow speeds. The double-layer

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 Table II The average of both single-layer samples, the average of the double-layer sample, the sum of the two single-layer samples, the absolute difference between the average single-layer sample's
 response ar ference bet layers

 Area of clock between the average of double layer sample's
 Area of clock between the layer sample's
 response ar ference bet layers

response and the double-layer sample's response, and finally, the difference between the double-layer response and the sum of the two layers

Airflow speed (km/h)	Avg of single-layer samples	Avg of double-layer sample	Sum of single-layer samples (Sample 1 + Sample 2)	(Single-+double- layer samples)	Double-layer – sum of single-layer samples
24 ± 1.5	105.429	130.857	210.857	25.429	80.00
21 ± 1.5	79.571	114.286	159.143	34.714	44.857
18 ± 1.5	75.429	99.714	150.857	24.286	51.143
15 ± 1.5	52.857	84.286	105.714	31.429	21.429
12 ± 1.5	43.714	79.143	87.429	35.429	8.286
9 ± 1.5	35.286	73.429	70.571	38.143	-2.857
6±1.5	18.857	64.857	37.714	45.00	-27.143

sample had greater voltage outputs than the sum of the 828 two individual sample's voltages until the speed reached 829 12 ± 1.5 m/s, after which the sum of the two samples con-830 tinued to increase compared to the double-layer sample. 831 The absolute difference between the two samples also 832 shows a steady decrease as the speed increases, which 833 again shows that at higher speeds, the use of two layers 834 instead of one becomes less efficient. 835

836 Conclusion

In this work, PVDF electrospun nanofibers with good mor-837 phology, optical activity, beta sheet content, and piezo activ-838 ity were fabricated at optimized experimental conditions. 839 The feeder-to-collector distance was varied from 12 cm to 840 15 cm, and the generated nanomats clearly demonstrated the 841 developmental stages of nanofiber membranes from a thin 842 cast-like film. Film-like behavior was exhibited by the 12 cm 843 electrospun membrane, and the increase in distance from 844 13 cm to 15 cm clearly showed the generation of fiber-like 845 nanostructures, with a final successful nanofiber mat genera-846 tion at the 15 cm distance. Appreciable optical activity was 847 observed using a UV-Vis spectrophotometer, and the result 848 clearly demonstrated a trade-off between piezoresponse 849 and transparency. The best piezo membrane exhibited the 850 851 lowest transmittance, and the transparent sample showed weak piezo activity. The beta sheet contents were calculated 852 and found to be in the range of 69% to 78.7%. Mechani-853 854 cal disturbances in the form of variable forces, frequency, and impulse loading on the most efficient sample generated 855 a maximum of 4.5 V potential, which clearly revealed the 856 857 potential of PVDF nanomats in the field of energy harvesting. The dependence of the piezo potential on the resistance 858 loads was explored using power density measurements, and 859 860 a Lorentzian curve-based graphical plot was obtained for the 15 cm sample with a maximum of 21 μ W/m², while the other 861 sheets exhibited random behavior. The possibility of apply-862 ing this efficient nanosheet in energy harvesting and sensing 863

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disciplines was studied by applying disturbances in the form 864 of mechanical shocks by variable momentum via loads and 865 varying airflow speeds. Energy harnessing and sensing 866 behavior of the nanomats was tested for single- and double-867 layer samples, and the piezo effect was found to be enhanced 868 with a double layer. A clear enhancement in airflow sens-869 ing efficiency was also shown by the wind-based studies for 870 the double-layer sample, paving the way for application in 871 complex systems for detecting gas leakages. Thus, a detailed 872 analysis based on electrospun PVDF-based nanosheets was 873 performed, and their potential in the field of ambient energy 874 harvesting and sensing was thoroughly studied. 875

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Data availabilityData are available based upon a request sent by email891to the corresponding author.892

Conflict of interestNo potential conflicts of interest was reported by893the authors.894

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